



# Annual Report 2014

MLZ is a cooperation between:

# On the complex H-bonding network in paravauxite, $\text{Fe}^{2+}\text{Al}_2(\text{PO}_4)_2(\text{OH})_2 \cdot 8 \text{H}_2\text{O}$

G. D. Gatta<sup>1</sup>, P. Vignola<sup>1,2</sup>, M. Meven<sup>3,4</sup>

<sup>1</sup>Dipartimento di Scienze della Terra, Università degli Studi di Milano, Milano, Italy

<sup>2</sup>CNR-Istituto per la Dinamica dei Processi Ambientali, Milano, Italy

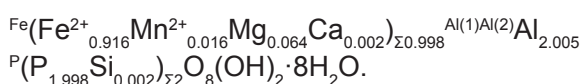
<sup>3</sup>Institut für Kristallographie, RWTH Aachen, Aachen, Germany

<sup>4</sup>Jülich Centre for Neutron Science (JCNS) at MLZ, Forschungszentrum Jülich GmbH, Garching, Germany

Phosphate minerals represent the major host for transition metals and  $\text{H}_2\text{O}$  in pegmatitic rocks, playing an essential geochemical role in the evolution processes of pegmatites. A good knowledge of their crystal chemistry is therefore necessary to better understand the genesis of pegmatites. Paravauxite is a mineral found in hydrothermal tin veins and granite pegmatites [1,2]. Its ideal chemical formula is  $\text{Fe}^{2+}\text{Al}_2(\text{PO}_4)_2(\text{OH})_2 \cdot 8\text{H}_2\text{O}$ . Its crystal structure was solved and refined by Baur [3] in 1969 on the basis of single-crystal X-ray diffraction data. This structure model appears to be consistent. However, due to the technical limitations of X-ray diffraction, the refinement only provided the isotropic displacement parameters, and the positions of nine independent proton sites were assigned but not refined. This led to a poor description of (the expected) complex H-bonding scheme in the paravauxite structure. In light of this, the crystal structure of a natural paravauxite was reinvestigated using electron microprobe analysis in wavelength dispersive mode (EPMA-WDS) and single-crystal neutron diffraction in an attempt to resolve these open questions.

## Looking into a gem stone

A gemmy, pale green, single crystal of paravauxite (up to 9 mm in length and 5 mm in diameter) from the Siglo Veinte Mine, Bolivia, was used in this study. The determination of the chemical composition was performed by EPMA-WDS analysis on a polished crystal using a Jeol JXA-8200 microprobe with the following result:



A single-crystal neutron diffraction experiment was

performed using the hot source (fast neutrons) single-crystal diffractometer HEiDi of the neutron source FRM II. The diffraction data were collected at 293 K with a wavelength of the incident beam of 1.1680(2) Å. The unit-cell parameters were refined on the basis of the 42 Bragg reflections (space group:  $P-1$ ,  $a = 5.240(6)$  Å,  $b = 10.567(7)$  Å,  $c = 6.698(9)$  Å,  $\alpha = 106.82(8)^\circ$ ,  $\beta = 110.77(9)^\circ$ ,  $\gamma = 72.23(9)^\circ$ ,  $V = 336.4(6)$  Å<sup>3</sup>). A total number of 4190 reflections were collected up to  $2\theta_{\text{max}} = 126.3^\circ$  and  $\sin(\Theta)/\lambda = 0.76/\text{Å}$ , respectively. The discrepancy factor for the symmetry related reflections (based on Friedel pairs) was  $R_{\text{int}} = 0.0442$ . The anisotropic structure refinement was then performed using the SHELX-97 software [4], starting from the atomic coordinates of Baur [3] without H sites. The structure refinement was conducted with: *a*) the neutron scattering length of iron at the octahedral Fe site and the scattering length of aluminum at the octahedral Al(1) and Al(2) sites, also refining their site occupancy factors (*s.o.f.*); *b*) the scattering length of phosphorous at the tetrahedral P site, with full occupancy; *c*) the scattering length of oxygen at the OP(1), OP(2), OP(3), OP(4), OH(5), OW(6), OW(7), OW(8) and OW(9), with full site occupancies. Then, a structure model was implemented with nine H sites, (*i.e.*, H(1), H(2), H(3), H(4), H(5), H(6), H(7), H(8) and H(9)) all at  $\sim 1$  Å from the respective O sites. Given such a model, convergence was rapidly achieved. However, H(4) and H(9) showed unrealistically large displacement parameters, if compared to those of the other H sites. Further refinement cycles were then conducted splitting the H(4) and H(9) sites into two mutually exclusive sub-sites (*i.e.*, H(4A) and H(4B), H(9A) and H(9B)) only 0.4–0.6 Å apart. Their *s.o.f.* were not restrained. With this configuration, the refined displacement parameters had realistic values, convergence was achieved and the variance-covariance matrix showed no significant correlation among the refined parameters. No peak larger than



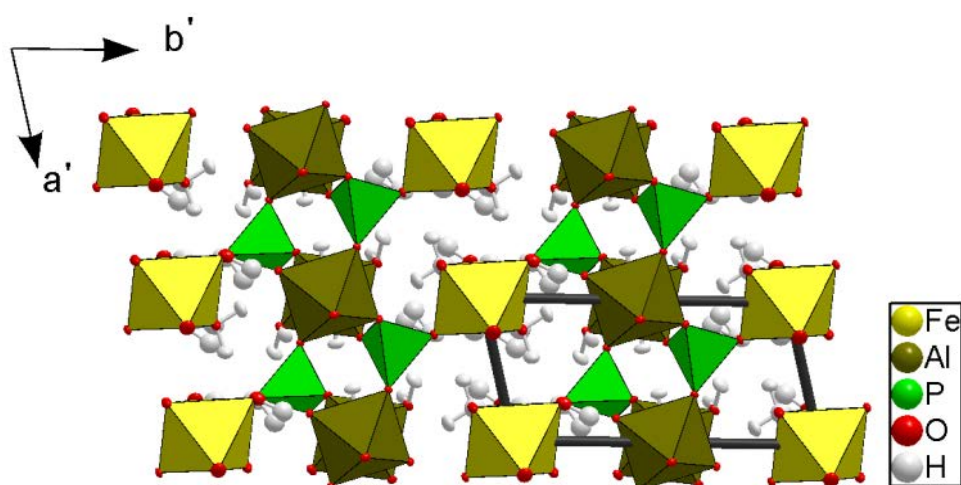


Figure 1: The crystal structure of paravauxite viewed down [001].

$\pm 1.3 \text{ fm}/\text{\AA}^3$  was present in the final difference-Fourier map of the nuclear density. The final statistical index  $R_1$  was 0.0495 for 194 refined parameters and 1678 unique reflections with  $F_o > 4\sigma(F_o)$ .

### Locating the hydrogen in paravauxite

This is the first study in which the crystal structure of paravauxite has been investigated on the basis of single-crystal neutron diffraction. Previous structure data available in the literature [3] are based on single-crystal X-ray diffraction. The structural refinement of this study confirms the former general structure model [3]. The structure of paravauxite is composed of chains of corner-sharing Al-octahedra, running along [001], linked by P-tetrahedra to form layers parallel to the *ac*-plane. These layers are connected by Fe-octahedra (Fig. 1). Two independent Al-octahedra (*i.e.*,  $\text{AlO}_4(\text{OH})_2$  and  $\text{AlO}_2(\text{OH})_2(\text{OH}_2)_2$ ), one independent Fe-octahedron (*i.e.*,  $\text{FeO}_4(\text{OH}_2)_2$ ) along with one independent  $\text{PO}_4$ -tetrahedron form the polyhedral “framework”, and at least one independent “zeolitic”  $\text{H}_2\text{O}$  lies in the cavities.

Using the neutron scattering length of iron at the Fe site, the refined occupancy factor is *s.o.f.* = 0.921(7). This virtual partial site occupancy reflects the multi-elemental population at the Fe site, as shown by the EPMA-WDS [*i.e.*, with minor fractions of Mg (0.064 *a.p.f.u.*) and Mn (0.016 *a.p.f.u.*)]. The Al(1) and Al(2) sites were found to be fully occupied by aluminum (with refined *s.o.f.* = 1.02(2) and 1.05(2), respectively). The *s.o.f.* of the subsites H(4A) and H(4B), and H(9A) and H(9B) were re-

finied without any restraint, and the sum [*s.o.f.*(H4A) + *s.o.f.*(H4B)] = 0.94(3) and [*s.o.f.*(H4A) + *s.o.f.*(H4B)] = 1.02(2) suggest full site occupancies within  $2\sigma$ . The structure model with the sub-sites H(4A) and H(4B), and H(9A) and H(9B) is the best fit to the observed intensity data (at 293 K), with realistic displacement parameters.

The complex H-bonding scheme in the paravauxite structure is now well defined, with twelve independent H-bonds. Some of the H-bonds appear to be stronger than others. The weaker are characterized by low O-H...O angular values (*i.e.*, 123–146°). Some H-bonds connect the Al-octahedra with the Fe-octahedra. The zeolitic  $\text{H}_2\text{O}$  molecule (*i.e.*, H(8)-OW(9)-H(9AB)) is connected via H-bonding to OP(1) (*i.e.*, the bridging oxygen between the Al(1)-octahedron and the P-tetrahedron), OP(3) (*i.e.*, the bridging oxygen between the Fe-octahedron and the P-tetrahedron) and OW(6) (*i.e.*, belonging to the Al(2)-octahedron). Further structural details are reported in [5].

- [1] S. G. Gordon, *Proc. Acad. Nat. Sci. Philadelphia* 75, 261 (1922).
- [2] S. G. Gordon, *Proc. Acad. Nat. Sci. Philadelphia* 96, 279 (1944).
- [3] W. H. Baur, *Neues Jahrbuch für Mineralogie Monatshefte* 1969 430 (1969).
- [4] G.M. Sheldrick, *Acta Crystallogr. Sect. A* 64, 112 (2008).
- [5] G. D. Gatta et al., *Mineralogical Magazine* 78, 841 (2014).

## Evaluation of Beam Time Proposals: Members of the Review Panels

Dr. Tamás Belgya  
Budapest Neutron Center, Budapest

Dr. Victor Bodnarchuk  
Joint Institute for Nuclear Research  
Frank Laboratory of Neutron Physics, Dubna

Prof. Dr. Jan Bonarski  
Polish Academy of Sciences  
Institute of Metallurgy and  
Materials Science, Kraków

Dr. Laszlo Bottyan  
Hungarian Academy of Sciences  
Institute for Particle and Nuclear Physics, Budapest

Prof. Roberto Brusa  
Università degli Studi di Trento  
Facoltà di Ingegneria, Dipartimento di Fisica, Trento

Prof. Dr. Roberto Caciuffo  
Institute for Transuranium Elements  
Joint Research Center, Karlsruhe

Dr. Monica Ceretti  
Université de Montpellier 2  
Institut Charles Gerhardt, Montpellier

Dr. Niels Bech Christensen  
Technical University of Denmark  
Institute of Physics, Roskilde

Dr. Pascale Deen  
European Spallation Source (ESS AB), Lund

Dr. Sabrina Disch  
University of Cologne,  
Department of Chemistry, Cologne

Prof. Dr. Stefan Egelhaaf  
Heinrich-Heine-Universität Düsseldorf  
Lehrstuhl für Physik der weichen Materie,  
Düsseldorf

Prof. Dr. Helmut Ehrenberg  
Karlsruher Institut für Technologie (KIT)  
Institut für Angewandte Materialien, Karlsruhe

Dr. Tom Fennell  
Paul Scherrer Institute  
Laboratories for Solid State Physics  
Neutron Scattering

Dr. Marie Thérèse Fernandez-Diaz  
Institut Laue-Langevin (ILL), Grenoble

Dr. Peter Fouquet  
Institut Laue-Langevin (ILL), Grenoble

Dr. Victoria Garcia-Sakai  
STFC Rutherford Appleton Laboratory, Didcot

Prof. Giacomo Diego Gatta  
Università degli Studi di Milano  
Dip. Scienze della Terra „Ardito Desio“, Milano

Prof. Dr. Rupert Gebhard  
Archäologische Staatssammlung München,  
Abt. Vorgeschichte, München



Figure 3: Members of the Review Panels from left to right (background): A. Radulecu, J. Neuhaus, O. Stockert, A. Ostermann, T. Nylander, K. Temst, A. Schneidewind, H. Ehrenberg, A. Senyshyn, W. Petry, S. Mattauch, C. Piochacz, A. Magerl, W. Sprengel, P. Schurtenberger. From left to right (foreground): F. Carsughi, P. Staron.

Dr. Jens Gibmeier  
Karlsruher Institut für Technologie (KIT),  
Institut für Angewandte Materialien, Karlsruhe

Dr. Béatrice Gillon  
Laboratoire Léon Brillouin  
CEA, Centre de Saclay

Dr. Arsène Goukassov  
Laboratoire Léon Brillouin  
CEA, Centre de Saclay

Dr. Christian Grünzweig  
Paul Scherrer Institute, Villigen

Prof. Dr. Ian William Hamley  
University of Reading  
Department of Chemistry, Reading

Dr. Thomas Hauss  
Helmholtz-Zentrum Berlin für Materialien  
und Energie GmbH, Berlin

Prof. Dr. Stephen Hayden  
University of Bristol  
HH Wills Physics Laboratory, Bristol

Dr. Arno Hiess  
European Spallation Source (ESS AB),  
Neutron Science Division, Lund

Dr. Klaudia Hradil  
Technische Universität Wien,  
Röntgenzentrum, Wien

Dr. Nikolay Kardjilov  
Helmholtz-Zentrum Berlin für Materialien  
und Energie GmbH, Berlin

Prof. Dr. Michel Kenzelmann  
Paul Scherrer Institute, Laboratories for Solid State  
Physics & Neutron Scattering, Villigen

Prof. Christian Krempaszky  
Technische Universität München,  
Fakultät für Maschinenwesen, München

Dr. Eberhard Lehmann  
Paul Scherrer Institute, Villigen

Prof. Dr. Martin Lerch  
Technische Universität Berlin,  
Institut für Chemie, Berlin

Dr. Dieter Lott  
Helmholtz-Zentrum Geesthacht GmbH,  
Institut für Materialforschung, Geesthacht

Prof. Dr. Andreas Magerl  
Universität Erlangen-Nürnberg,  
Kristallographie und Strukturphysik, Erlangen

Dr. Andreas Michels  
Université de Luxembourg,  
Faculté des Sciences, de la Technologie  
et de la Communication, Luxembourg

Dr. Bert Nickel,  
Ludwig Maximilians-Universität München,  
Fakultät für Physik, München

Prof. Dr. Tommy Nylander  
Lund University, Physical Chemistry, Lund

Prof. Dr. Luigi Paduano  
University of Naples "Federico II",  
Chemistry Department, Naples

Prof. Dr. Catherine Pappas  
Delft University of Technology, Delft

Prof. Dr. Oskar Paris  
Montanuniversität Leoben, Leoben

Prof. Dr. Wolfgang Paul  
Martin-Luther-Universität Halle-Wittenberg,  
Institut für Physik, Halle

Dr. Simon Redfern University of Cambridge,  
Department of Earth Sciences, Cambridge

Prof. Dr. Günther Redhammer Universität Salzburg,  
Materialforschung und Physik, Salzburg

Dr. Matthias Rossbach  
Forschungszentrum Jülich GmbH, Jülich

Dr. Margarita Russina  
Helmholtz-Zentrum Berlin GmbH,  
Institut Weiche Materie und Funktionale Materialien,  
Berlin

Prof. Dr. Michael Sattler  
Technische Universität München,  
Department Chemie, München

Dr. Harald Schmidt  
Technische Universität Clausthal,  
Institut für Metallurgie, Clausthal

Prof. Dr. Andreas Schönhals  
Bundesanstalt für Materialforschung und -prüfung,  
Berlin

Prof. Dr. Peter Schurtenberger University of Lund,  
Physical Chemistry 1, Lund

Dr. Torsten Soldner  
Institut Laue-Langevin (ILL), Grenoble

Prof. Dr. Wolfgang Sprengel  
Technische Universität Graz,  
Institut für Materialphysik, Graz

Dr. Jochen Stahn  
ETH Zürich and Paul Scherrer Institute, Villigen

Dr. Peter Staron  
Helmholtz-Zentrum Geesthacht GmbH  
Institute of Materials Research, Geesthacht

Dr. Paul Steffens  
Institut Laue-Langevin (ILL), Grenoble

Dr. Oliver Stockert  
Max-Planck-Institut für Chemische Physik fester  
Stoffe Dresden, Dresden

Dr. Susana Teixeira  
Institut Laue-Langevin (ILL), Grenoble

Prof. Kristiaan Temst  
Katholieke Universiteit Leuven,  
Nuclear & Radiation Physics Section, Leuven





Figure 4: First discussions among the referees during the welcome buffet for the review.

Prof. Dr. Katharina Theis-Broehl  
Hochschule Bremenhaven, Bremenhaven

Prof. Dr. Thomas Thurn-Albrecht  
Martin-Luther-Universität Halle-Wittenberg,  
Experimentelle Polymerphysik, Halle

Prof. Dr. Tobias Unruh  
Universität Erlangen-Nürnberg,  
Kristallographie und Strukturphysik, Erlangen

Dr. Lambert van Eijck  
Delft University of Technology,  
Department of Radiation, Radionuclides  
and Reactors, Delft

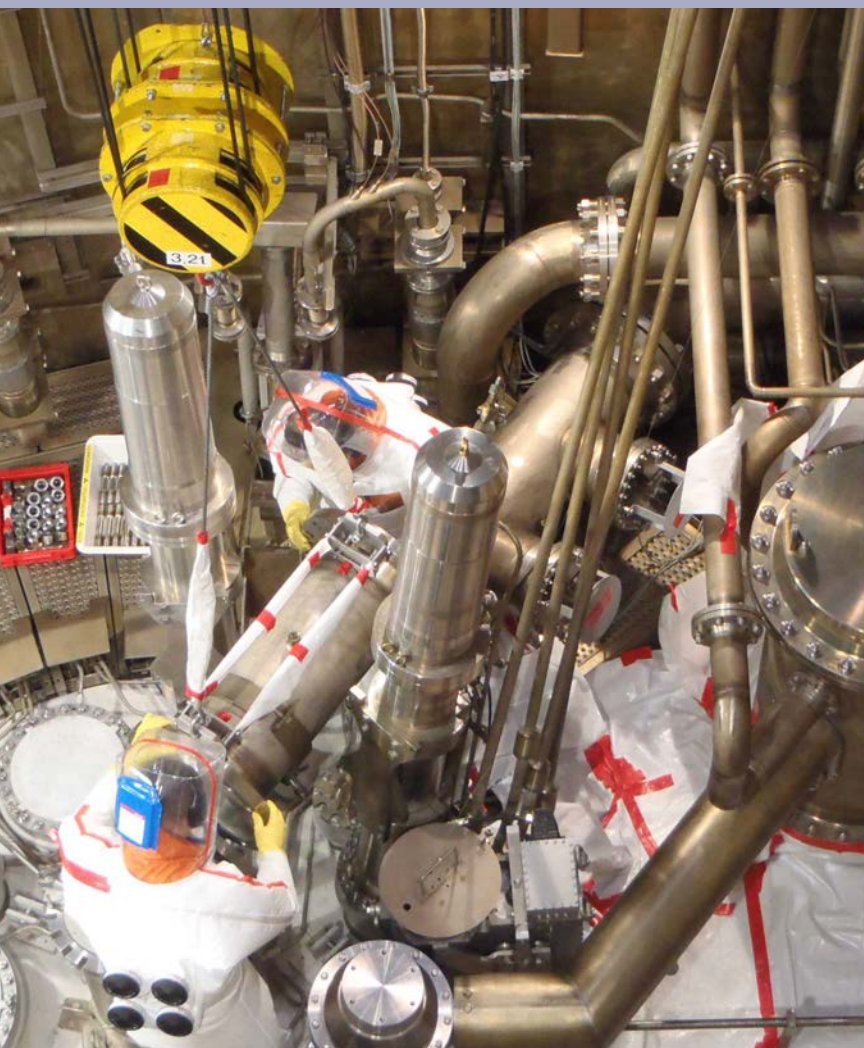
Prof. Dr. Regine von Klitzing  
Technische Universität Berlin,  
Institut für Chemie, Stranski-Laboratorium  
für Physikalische und Theoretische Chemie, Berlin

Dr. Martin Weik  
Institut de Biologie Structurale, Grenoble

Dr. Andrew Wildes  
Institut Laue-Langevin (ILL), Grenoble

Dr. Robert Wimpory  
Helmholtz-Zentrum Berlin für Materialien und  
Energie GmbH, Berlin





**Heinz Maier-Leibnitz Zentrum (MLZ)**

[www.mlz-garching.de](http://www.mlz-garching.de)

DOI: 10.14459/2015md1239870